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Enhancing TiO₂ Photoelectrode Performance: The Role of La Doping Using the Sol-Gel Process

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Abstract

Due to growing interest in renewable energy technologies, dye-sensitized solar cells (DSSCs) have emerged as promising alternatives to conventional photovoltaics. This study explores the enhancement of titanium dioxide (TiO₂) photoelectrodes through doping with lanthanum (La), a rare earth element known for its ability to modify semiconductors' structural and electronic properties. La-doped TiO₂ was synthesized using the sol-gel method with La concentrations ranging from 0 to 4 mol%. The resulting films were deposited onto substrates and sintered at 600 °C for 1 hour. Characterization techniques included X-ray diffraction (XRD), scanning electron microscopy with energy-dispersive spectroscopy (SEM-EDS), UV-Vis spectrophotometry, and Fourier-transform infrared spectroscopy (FTIR). XRD confirmed the anatase phase of TiO₂ and the appearance of La₂O₃ phases with doping. The smallest crystallite size (8.20 nm) and lowest bandgap energy (3.31 eV) were achieved at 1 mol% La, compared to 3.52 eV for undoped TiO₂. SEM-EDS indicated uniform La distribution, while FTIR revealed changes in surface chemistry. These results suggest that La doping, particularly at 1 mol%, can effectively enhance the optical and structural properties of TiO₂, making it a promising candidate for improved DSSC performance.



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1. Introduction

Environmental issues and the depletion of energy resources are among the most pressing challenges facing the world in the next 50 years [1]. One of the main causes is the increasing consumption of energy, which has led to a rapid decline in petroleum reserves [2]. As a result, researchers are actively seeking alternative solutions, with renewable energy technologies gaining significant attention [3]. Among various renewable energy sources,

solar energy stands out due to its abundance and potential to be converted into electricity [4].

To harness solar energy efficiently, cost-effective and high-performance photovoltaic devices are essential [5]. Over the past few decades, three generations of photovoltaic technologies have been developed, with Dye-Sensitized Solar Cells (DSSCs) emerging as a promising third-generation option. First introduced in 1991, DSSCs operate based on the photovoltaic principle,

where light energy excites electrons on the cell's surface [4, 6].

A DSSC consists of several key components: a substrate, a photoelectrode, a dye, an electrolyte, and a counter electrode (commonly made of carbon) [7]. The photoelectrode plays a crucial role in converting light energy into electrical energy. Typically, oxide semiconductor materials are used for photoelectrodes due to their stability against photo corrosion and their relatively low bandgap energy, which enables efficient absorption of photons within the visible light spectrum [8].

Titanium dioxide (TiO₂) is one of the most commonly used semiconductors in DSSCs because it offers several advantages such as strong photo-oxidizing power, photocatalytic activity, affordability, and availability. However, TiO₂ also has some limitations, notably its relatively large bandgap, which leads to a higher rate of electron-hole recombination [6]. To overcome these drawbacks, doping TiO₂ with other elements has been explored. Doping can modify the base material's morphology, structure, electrical, and optical properties [6, 9].

One such dopant is lanthanum (La), which has an electron configuration of 5d¹6s² [10]. It typically forms a stable La³⁺ ion by losing three electrons in high-energy reactions. Due to its larger ionic radius, La³⁺ can form a secondary phase within TiO₂, potentially creating a solid solution. La doping has been shown to improve the optical response of TiO₂ and reduce its bandgap, enhancing its performance in DSSCs.

Previous research by Shakir et al. [11] demonstrated that Mg-doping in TiO₂ reduced the bandgap from 3.2 eV to 2.8 eV. Changes in Mg doping levels also affected the crystal size of TiO₂, decreasing from 53.92 nm to 49.23 nm [12]. Additionally, annealing temperature influences the bandgap; higher annealing temperatures tend to reduce it further [13]. Similarly, La doping has been shown to lower the TiO₂ bandgap from 2.98 eV to 2.75 eV [14].

In this study, we investigate the effect of lanthanum (La) doping on the structural and optical properties of TiO₂ using the sol-gel method. Specifically, we examine how varying concentrations of La influence the crystallinity, bandgap, and overall performance-related characteristics of TiO₂ for potential application in DSSCs. Although previous studies have demonstrated the benefits of La doping, a detailed understanding of how different doping levels impact TiO₂ properties remains limited. Therefore, the aim of this research is to provide a comprehensive analysis of La-doped TiO₂ to optimize its

potential as a photoelectrode material in dye-sensitized solar cells.

2. Materials and Methods

2.1. Synthesis of Pure and La-Doped TiO₂

The pure and La-doped TiO₂ samples were doping using the sol-gel method. The primary materials used included Titanium (IV) isopropoxide (Sigma Aldrich, 97%), Magnesium Acetate Tetrahydrate (Sigma Aldrich, 99%), and Lanthanum (III) Acetate Hydrate (Sigma Aldrich, 98%). The solvents utilized were isopropanol, distilled water, hydrochloric acid (HCl), and ethanol.

Initially, 4 mol% of Titanium (IV) isopropoxide was mixed with isopropanol and distilled water in a volume ratio of 50:5. The mixture was stirred until it turned white, indicating the beginning of hydrolysis. Subsequently, Lanthanum (III) Acetate Hydrate was added in varying concentrations (0 mol%, 1 mol%, 2 mol%, 3 mol%, and 4 mol%), followed by adding 1 mL of HCl to catalyze the reaction. The solution was stirred for 2 hours, and the gel was allowed overnight at room temperature. The resulting gel was dried in an oven, washed twice with ethanol, and further dried at 60°C. Finally, the samples were sintered at 500°C for 5 hours to obtain the desired crystalline form.

2.2. Preparation of TiO₂ Paste and Film Deposition

The TiO₂ paste was prepared by mixing the synthesized pure and La-doped TiO₂ powder with 0.76 mL of acetic acid, 10 mL of ethanol, and two drops of Triton X-100. The mixture was stirred until a homogeneous paste was formed. The paste was then deposited onto a 2.5 × 2.5 cm conductive glass substrate using the doctor blade method. After deposition, the films were sintered at 600°C for 1 hour to enhance adhesion and crystallinity.

2.3. Structural Characterization (XRD)

X-ray diffraction (XRD) analysis was carried out using a Shimadzu D-6000 diffractometer with Cu-K α radiation. Scanning was performed at a rate of 4°/min over a 2 θ range of 5° to 80° to determine the crystal structure and phase composition. The average grain size (D) was calculated using the Scherrer equation (Equation 1) [15]:

$$D = \frac{0.9\lambda}{B \cos \theta} \quad (1)$$

where D is the average grain size in angstroms (Å), λ is the X-ray wavelength, B is the full width at half maximum (FWHM) of the diffraction peak, and θ is the Bragg angle corresponding to the peak.

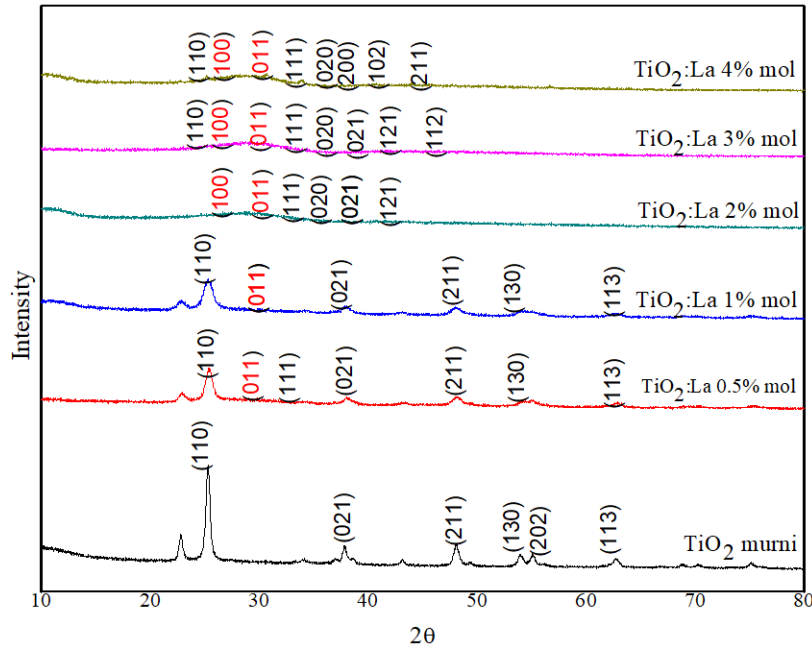


Figure 1. X-ray diffraction pattern of La-doped TiO₂ layer.

Table 1. Crystal size of TiO₂:La layers.

No	Sample Name	2θ (°)	θ (°)	FWHM	Crystal Size (nm)	Phase
1	TiO ₂ (pure)	25.2580	12.6290	0.4512	19.08	TiO ₂
2	TiO ₂ :La (0.5 mol%)	25.3574	12.6787	0.8311	10.34	TiO ₂
3	TiO ₂ :La (1 mol%)	25.2880	12.6440	1.0475	8.20	TiO ₂

2.4. Optical Property Analysis (UV-Vis)

The optical properties of the samples were analyzed using UV-Vis spectrophotometry. The photon energy (*E*) corresponding to the absorption wavelength was calculated using Planck’s equation (Equation 2) [16]:

$$E = \frac{hc}{\lambda} \tag{2}$$

In this equation, *E* represents photon energy (eV), *h* is Planck’s constant (6.626 × 10⁻³⁴ J·s), *c* is the speed of light (3 × 10⁸ m/s), and *λ* is the wavelength of the incident light (m).

2.5. Functional Group Analysis (FTIR)

FTIR spectroscopy was conducted using a Shimadzu FTIR spectrometer [17] to identify functional groups present in the samples after doping. The spectra were recorded in the 4000–400 cm⁻¹ range using potassium bromide (KBr) pellets, with a wavenumber resolution of 1 cm⁻¹.

3. Results and Discussion

To identify the structure and crystal size of the TiO₂:La layers in each sample, a qualitative analysis was conducted using X-ray diffraction (XRD) with a Shimadzu instrument, employing CuKα radiation (λ = 1.54060 Å). The applied voltage was 40 kV, with a current of 30 mA.

The scanning method was performed in a step mode with a data interval of 0.02°, and the detected diffraction angle (θ) ranged from 10° to 80°. Based on the XRD images and data, phase identification was done by analyzing the 2θ angles, lattice spacing (*d*), intensity (*I*/*I*₀), phase composition, and crystal structure. The identification process matched the obtained data with the PDF-ICDD database, using the 2θ angle as the primary reference.

Figure 1 shows the XRD spectrum of the TiO₂:La layers, revealing the presence of TiO₂ phases at 2θ = 25.26°, 48.02°, and 42.6°, while La₂O₃ phases appeared at 2θ = 25.8° and 30.56°. The most dominant diffraction peaks in the La-doped TiO₂ layers corresponded to TiO₂ in pure TiO₂ and in 0.5 mol% and 1 mol% La-doped TiO₂ samples, indicating the formation of a crystalline structure. However, the diffraction peaks showed a decrease in crystallinity for TiO₂:La samples with 2 mol%, 3 mol%, and 4 mol% doping. These predominant diffraction peaks in these samples corresponded to La₂O₃, suggesting that as the La content increased, the crystallinity of the semiconductor material was altered.

Table 1 shows the crystal size of the TiO₂:La layers. The calculations indicate that La doping reduces the crystal size of TiO₂. The largest crystal size was observed in the pure TiO₂ sample, measuring 19.08 nm. This is attributed to the small Full Width at Half Maximum (FWHM) value

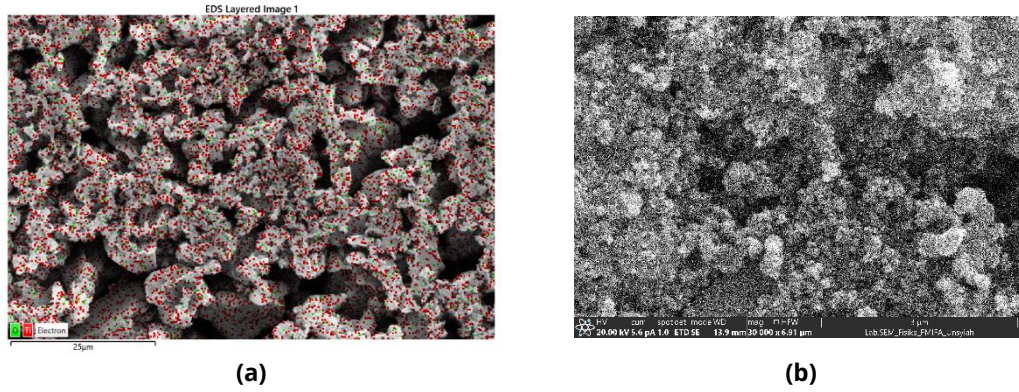


Figure 2. The SEM images of the TiO₂:La layers: (a) pure TiO₂ and (b) TiO₂:La with 1 mol% doping.

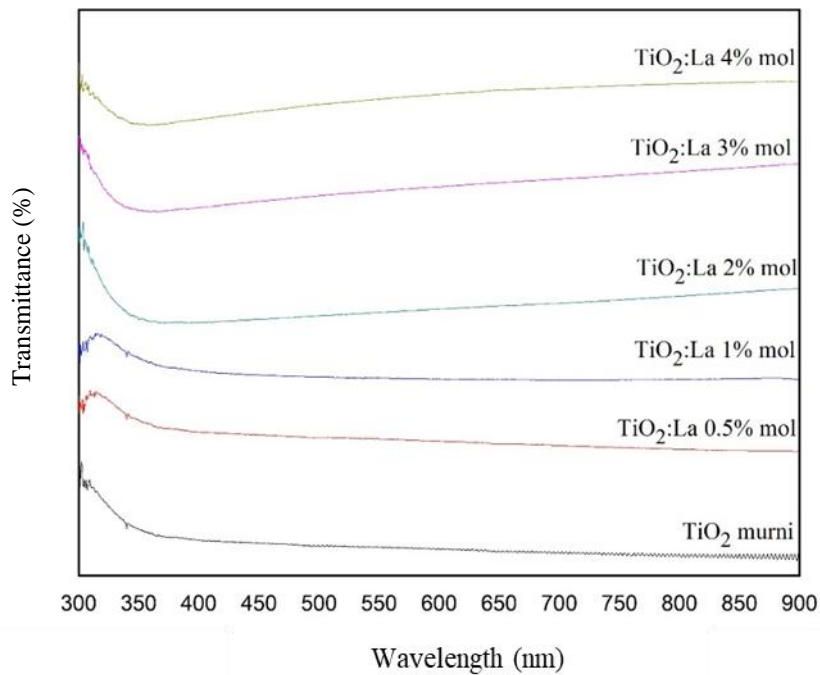


Figure 3. The transmittance graph of the La-doped TiO₂ layers.

and the narrow diffraction peaks. However, after adding 0.5 mol% and 1 mol% La doping, the TiO₂ crystal size decreased to 10.34 nm and 8.20 nm, respectively.

The results clearly show that La doping significantly affects the crystal size of TiO₂. However, at higher La doping concentrations (2 mol%, 3 mol%, and 4 mol%), the degree of crystallinity decreased, affecting the FWHM values. For these concentrations, the highest intensity diffraction peaks resulted in an FWHM value of zero, meaning the crystal size calculations had to be based on very low-intensity peaks. The zero FWHM value suggests that the structure of the material became amorphous, which was influenced by the increasing La concentration.

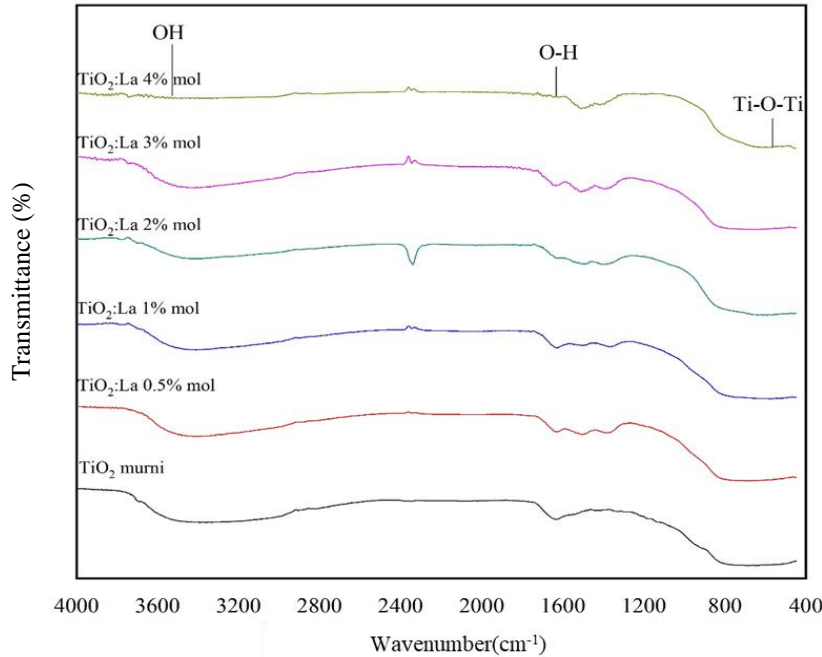
Figure 2 presents SEM images showing the surface morphology of TiO₂ layers: (a) pure TiO₂ and (b) TiO₂ doped with 1 mol% La. In Figure 2a, the pure TiO₂ exhibits a rough and porous structure with densely packed particles. The overlaid EDS mapping (red dots) indicates elemental distribution, suggesting a uniform

composition. In contrast, Figure 2b, representing La-doped TiO₂, shows a more refined and compact surface morphology, with slightly smaller and more uniformly distributed particles. This suggests that La doping influences the growth and distribution of TiO₂ grains, contributing to the formation of a more homogeneous surface, which is beneficial for photoelectrode applications in DSSCs.

The transmittance spectrum of the La-doped TiO₂ layers is presented in Figure 3. The obtained transmittance curves indicate optical absorption in the wavelength range of 330–450 nm for all samples, including pure TiO₂ and TiO₂ doped with 0.5 mol%, 1 mol%, 2 mol%, 3 mol%, and 4 mol% La. The peak transmittance of pure TiO₂ occurred at 352 nm. After doping with 1 mol% La, the transmittance peak shifted to 374 nm, indicating increased absorption towards longer wavelengths (redshift). However, for 4 mol% La-doped TiO₂, the transmittance peak shifted back to 349.5 nm, suggesting

Table 2. Bandgap energy of TiO₂:La layers.

Sample	Cutoff Wavelength (λ_0) (nm)	Bandgap Energy (eV)
TiO ₂ (pure)	352	3.52
TiO ₂ :La (0.5 mol%)	369	3.36
TiO ₂ :La (1 mol%)	374	3.31
TiO ₂ :La (2 mol%)	360.5	3.44
TiO ₂ :La (3 mol%)	347	3.57
TiO ₂ :La (4 mol%)	349.5	3.55

**Figure 4.** The FTIR spectrum of La-doped TiO₂ layers.

that excessive La doping reduces transmittance. These results indicate that as the La concentration increases, the transmittance value tends to decrease, shifting the absorption edge and influencing the optical properties of TiO₂. This suggests that La doping can effectively tune the optical behavior of TiO₂, which is beneficial for applications such as DSSCs.

As a photocatalyst, which is a material that accelerates light-induced reactions, TiO₂ has a semiconductor structure with an electronic structure characterized by a filled valence band and an empty conduction band. The two bands are separated by a band gap energy of 3.2 eV for the anatase phase and 3.0 eV for the rutile phase. TiO₂ in the anatase phase has a higher activity compared to the rutile phase. In the anatase phase, TiO₂ has a larger surface area and smaller size compared to rutile.

Table 2 shows that variations in La doping concentration significantly affect the bandgap energy of TiO₂. The bandgap of pure TiO₂ was measured at 3.52 eV. After doping with 1 mol% La, the bandgap decreased to 3.31 eV, indicating improved light absorption and potential enhancement in DSSC performance. The bandgap reduction in La-doped TiO₂ is more pronounced than Mg-

doped TiO₂, as La has a lower intrinsic bandgap than Mg. However, at higher La doping concentrations (3 mol% and 4 mol%), the bandgap increased to 3.57 eV and 3.55 eV, respectively. This increase is due to the excessive La content, which disrupts the TiO₂ crystal structure and reduces its crystallinity.

Reducing the bandgap through doping can minimize premature electron recombination during the excitation process, making La-doped TiO₂ a promising material for use as a photoelectrode in DSSCs. However, excessive La doping may negatively impact the material's electronic properties.

Figure 4 presents the FTIR spectrum of La-doped TiO₂ layers. The spectrum reveals the Ti-O-Ti polymer chain at 530 cm⁻¹, indicating the presence of TiO₂ bonding structures. The O-H vibrational mode is observed at 3414 cm⁻¹ and 1635 cm⁻¹, suggesting hydroxyl (-OH) group vibrations commonly associated with surface moisture or absorbed water. Incorporating La³⁺ ions into the TiO₂ matrix influences the adsorption of H₂O molecules on the material's surface. Despite testing six different La doping concentrations, the FTIR spectra showed no significant differences in peak positions. However, the addition of La

affected the H₂O-related functional groups, indicating that La doping alters the surface chemistry of TiO₂ (Huang et al., 2017). This surface modification could potentially enhance the photocatalytic activity and DSSC performance of La-doped TiO₂ by improving dye adsorption and charge transport.

Based on the crystal structure analysis, adding La doping alters the crystal structure of TiO₂ and reduces its grain size. A smaller grain size enhances the dye adsorption capability of the La-doped TiO₂ layer, which is beneficial for DSSC applications. From an optical perspective, La doping also reduces the energy gap of TiO₂. A lower bandgap facilitates electron excitation from the valence band to the conduction band, allowing more sunlight absorption by the La-doped TiO₂ layer. The experimental results indicate that La-doped TiO₂ has significant potential as a photoelectrode material for DSSCs, as it improves both optical and structural properties, leading to enhanced solar energy conversion efficiency. This includes categorizing the DSSC manufacturing process and comparing it to solar cell applications [18].

4. Conclusions

Based on the results of this study, it can be concluded that La doping enhances the crystallinity of TiO₂ compared to the undoped sample. The smallest crystallite size, 8.20 nm, was observed at a La concentration of 1 mol%. Surface morphology analysis showed a homogeneous distribution, and SEM-EDS results indicated that the atomic percentage of Ti decreased with increasing La concentration, suggesting successful substitution of Ti by La in the TiO₂ lattice. The lowest energy bandgap, 3.31 eV, was also achieved at 1 mol% La doping, confirming that La incorporation effectively reduces the bandgap of TiO₂. FTIR analysis revealed characteristic Ti–O–Ti and Ti–O bonds at 530 cm⁻¹ and 789 cm⁻¹, respectively, along with increased H₂O absorption in La-doped samples. TiO₂ doped with 1 mol% La shows the most favorable structural and optical properties, indicating its strong potential as a photoelectrode material for DSSC applications.

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